

# Coherent Resonant Tunneling through Double Metallic Quantum Well States

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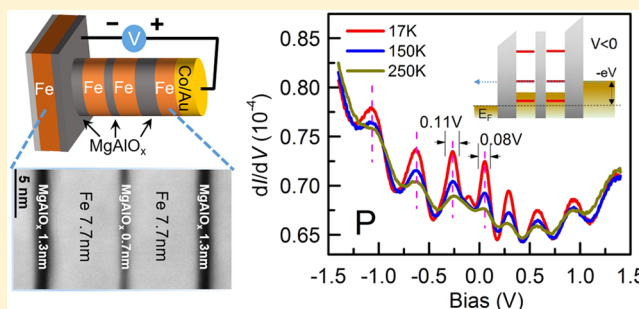
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## Supporting Information

**ABSTRACT:** Study of resonant tunneling through multi-metallic quantum well (QW) structure is not only important for the fundamental understanding of quantum transport but also for the great potential to generate advanced functionalities of spintronic devices. However, it remains challenging to engineer such a structure due to the short electron phase coherence length in metallic QW system. Here, we demonstrate the successful fabrication of double-QW structure in a single fully epitaxial magnetic tunnel junction (MTJ) heterostructure, where two Fe QW layers are sandwiched between three MgAlO<sub>x</sub> tunnel barriers. We show clear evidence of the coherent resonant tunneling through the discrete QW states in the two QWs. The coherent resonant tunneling condition is fulfilled only when the middle barrier between the two QWs is thin enough and available QW states are present simultaneously in both QWs under a certain bias. Compared to the single QW structure, the resonant tunneling in double-QW MTJ produces strong conductivity oscillations with much narrower peak width (about half) owing to the enhanced energy filtering effect. This study presents a comprehensive understanding of the resonant tunneling mechanism in MTJ with multiple QWs, which is essential for future development of new spintronic devices operating in the quantum tunneling regime.

**KEYWORDS:** Quantum well, magnetic tunnel junction, resonant tunneling, magnetoresistance



Resonant tunneling in quantum well (QW) structures has been extensively studied because of its importance in the field of nanoelectronic science and technology.<sup>1</sup> Recently, in the metallic QW system the combination of the tunneling magnetoresistance (TMR)<sup>2,3</sup> effect with resonant tunneling through metallic QW states in magnetic tunnel junctions (MTJs) has triggered considerable interest in the new functionality of spintronic devices.<sup>4–8</sup> In these structures, the discrete spin-dependent QW states can be formed in the middle metallic QW layer sandwiched between two barriers. The QW potential barrier can be formed either by metallic layer using the symmetry-dependent Bloch state orbital band structure,<sup>9–12</sup> or by a double oxide tunneling barriers with a much higher barrier height for better electron confinement.<sup>13–15</sup> Because the QW states are often formed in the

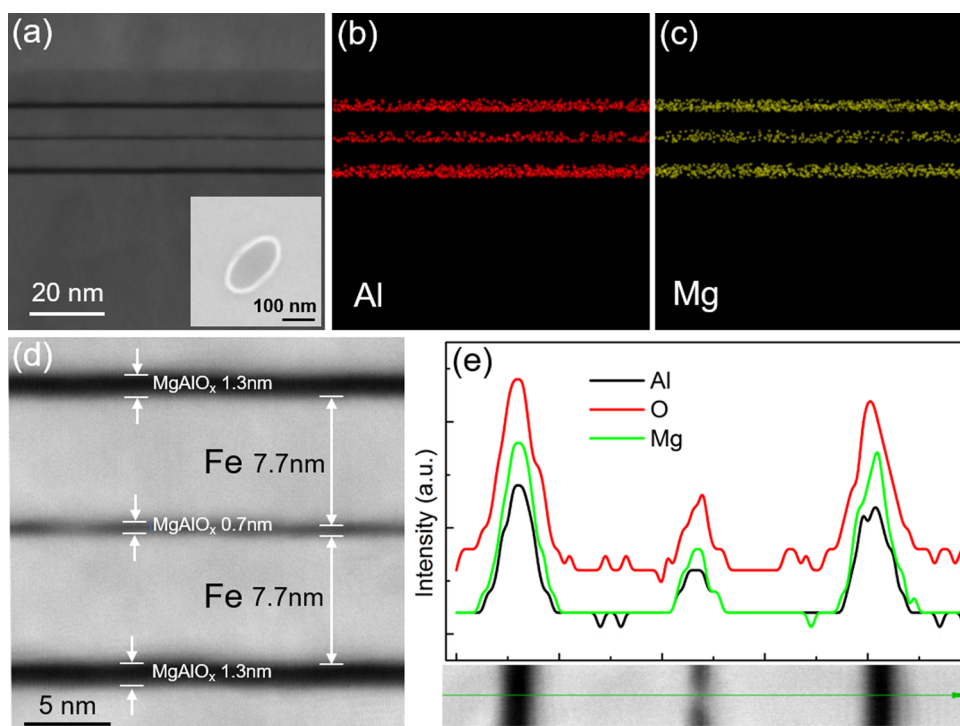
majority spin channel,<sup>4,8</sup> the conductance in parallel magnetic configuration can be enhanced when the injector Fermi level is aligned with the QW state at resonance, resulting in an enhanced TMR ratio.<sup>4</sup> A record high resonant TMR ( $\approx 4000\%$ ) has been reported for a double barrier MTJ measured with point-contact technique at 4.2 K.<sup>16</sup>

In semiconductor materials, multi-QW structures have already been extensively studied,<sup>17</sup> such as resonant-tunneling diode,<sup>18</sup> multi-quantum well light-emitting diode and laser,<sup>19</sup> and so forth. When the thickness of the barrier and QW is thin enough, the multi-QWs form the superlattice structure and the

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**Figure 1.** (a) Dark-field STEM image of the double QW MTJ. Inset: SEM top view of elliptical shape of nanopillar junction. (b,c) EDS mapping of Al and Mg elements. (d) Enlarged STEM image in the double QW region. (e) Profile of different elements and corresponding STEM image.

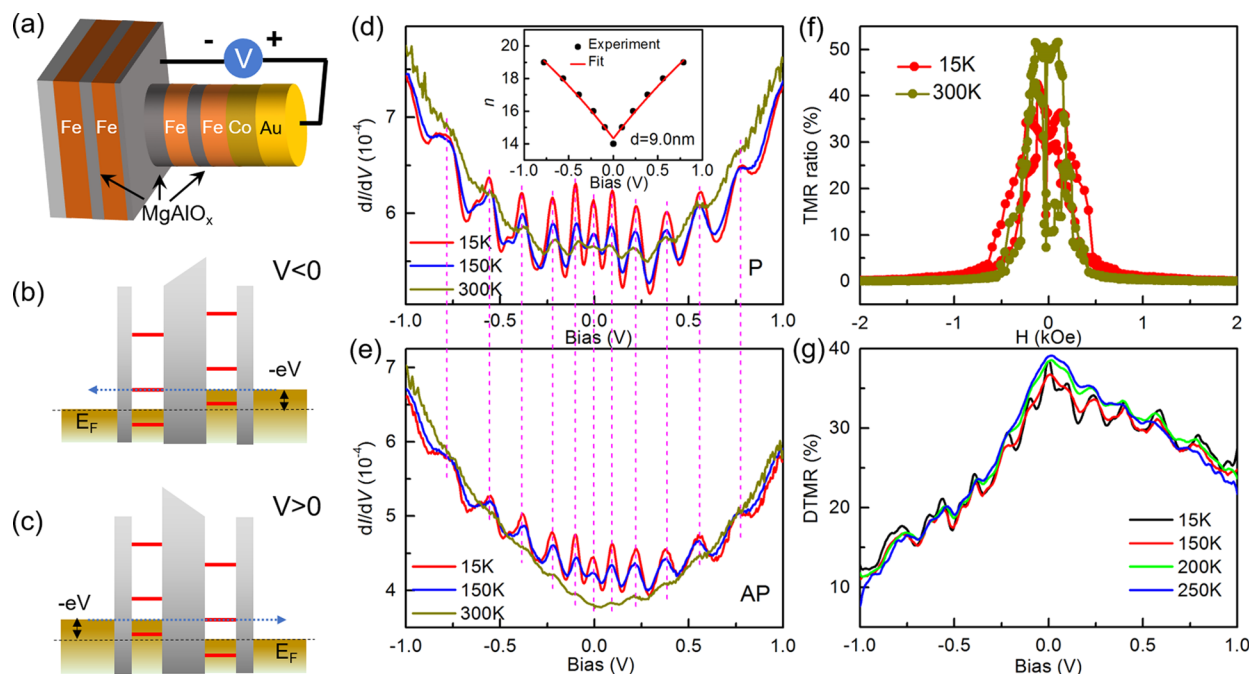
QW states in different wells can hybridize with each other to create new sub-band structure,<sup>20,21</sup> which gives rise to many novel features for electronic and optoelectronic properties. However, up to now no experimental results have been reported for the metallic multi-QW structures. The general reason is due to the difficulty to keep a good electron phase coherence when resonant tunneling through multi-metallic QWs. The decoherence process originated from the interface roughness,<sup>22</sup> or the inelastic scattering in the QW and at the interface,<sup>23</sup> can absorb and re-emit electrons losing their phase information. In addition, the small electron mean-free path in the metals (Fe: 1.5 nm)<sup>24</sup> also results in a rapid energy thermalization due to the electron–electron or electron–phonon scattering, which prevents electrons dwelling on the QW states before tunneling into a second QW. Those are the reason why elaborating efficient multi-QW structures is very challenging.

In our previous work,<sup>25</sup> by using MgAlO<sub>x</sub> as the barrier<sup>26–29</sup> to form the QW, we demonstrated the long-range phase coherence in thick Fe QW up to 12 nm, which was attributed to minimizing the interfacial strain at the Fe/MgAlO<sub>x</sub> interface due to the small lattice mismatch between Fe and MgAlO<sub>x</sub>.<sup>30,31</sup> In this work, by taking advantage of the long-range phase coherence in Fe/MgAlO<sub>x</sub> QW structure, we have successfully fabricated fully epitaxial double-QW MTJs with three MgAlO<sub>x</sub> tunnel barriers. The resonant tunneling through two QW states in a specifically designed double-QW structure has been evidenced by magneto-transport measurements. The understanding of the resonant tunneling mechanism in multi-QW MTJ provides efficient ways to engineer new functionalities for quantum resonant tunneling spintronic devices.

**Results and Discussion.** The single and double quantum-well (SQW and DQW) samples, namely MgO substrate//MgO (10 nm)/Fe (45 nm)/SQW or DQW/Fe (10 nm)/Co (10 nm)/Au (10 nm), were grown in a molecular beam epitaxy (MBE) chamber. Three types of DQW structures were

designed in this study with the stack structures from bottom to top as the following: Sample I, MgAlO<sub>x</sub>(3 MLs)/Fe(9 nm)/MgAlO<sub>x</sub>(6 MLs, etching stop layer)/Fe(9 nm)/MgAlO<sub>x</sub>(3 MLs); Sample II, MgAlO<sub>x</sub>(6 MLs, etching stop layer)/Fe(9 nm)/MgAlO<sub>x</sub>(6 MLs)/Fe(7.7 nm)/MgAlO<sub>x</sub>(6 MLs); and Sample III, MgAlO<sub>x</sub>(6 MLs, etching stop layer)/Fe(7.7 nm)/MgAlO<sub>x</sub>(3 MLs)/Fe(7.7 nm)/MgAlO<sub>x</sub>(6 MLs). One SQW sample (Sample IV) was grown to compare with the DQW Sample III with the structure MgAlO<sub>x</sub>(6 MLs)/Fe(7.7 nm)/MgAlO<sub>x</sub>(6 MLs, etching stop layer). Here, ML indicates atomic monolayer and the thickness of barrier was controlled by the high-energy electron diffraction (RHEED) intensity oscillation recorded during growth. The detailed condition can be found in [Methods](#) and [Supporting Information S1](#). After elaboration, E-beam lithography was used to form the nanopillar junctions with elliptical shape of 120 × 240 nm<sup>2</sup> (shown in the inset of [Figure 1a](#)). During the fabrication of nanopillar structure for SQW or DQW MTJ, the ion milling etching was stopped in a certain layer named “stop layer”. Please see in [Supporting Information S2](#) the effect of the position of the stop layer. For the two-terminal magneto-transport measurements, the negative bias voltage corresponds to the injection of electrons from top to bottom electrodes.

Scanning transmission electron microscope (STEM) characterizations were performed to verify the structure of the DQW MTJs. [Figure 1a,d](#) shows dark-field STEM images for Sample III in a large and magnified scale, respectively. In [Figure 1a](#), one can observe sharp interfaces between MgAlO<sub>x</sub> barriers (in black contrast) and Fe layers (in gray contrast). The QW and barrier layers appear quite flat and continuous, indicating a high quality of the triple-barrier MTJ structure. The magnified-scale image further confirms that the two Fe QW layers have an identical thickness of 7.7 nm. Energy-dispersive X-ray (EDS) spectroscopy has been performed to examine the chemical composition of the barriers. [Figure 1b,c](#)



**Figure 2.** (a) Schematics of the double QW MTJ with core structure of Fe/MgAlO<sub>x</sub>(3 ML)/Fe/MgAlO<sub>x</sub>(6 ML)/Fe/MgAlO<sub>x</sub>(3 ML)/Fe and the setup of measurement. (b,c) Schematics of resonant tunneling in the sample with bias  $V < 0$  and  $V > 0$ , respectively. (d,e) Bias dependence of differential conductance for the junction in magnetic parallel and antiparallel states, respectively. Inset in (d) shows the energy positions of QW states and the fitting by PAM method. (f) TMR– $H$  curves of the junction at  $T = 15$  K and 300 K. (g) Bias dependence of differential TMR ratio at different temperatures.

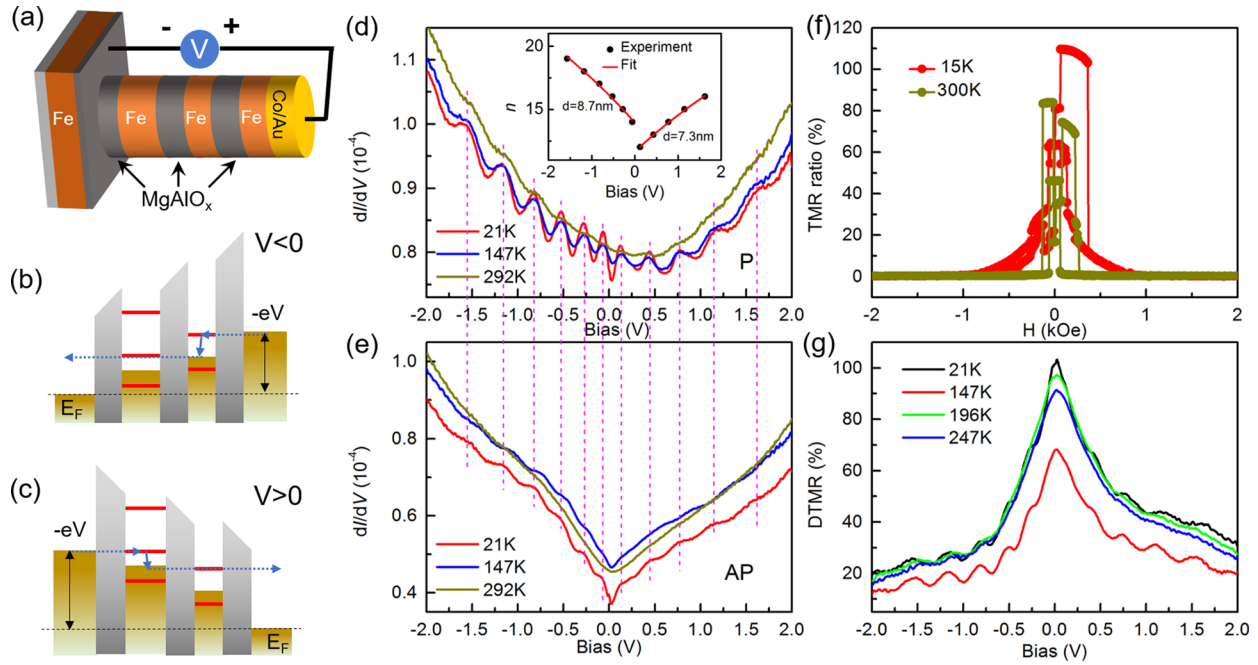
shows, respectively, the mapping of Al and Mg elements, indicating their homogeneous distribution in the barriers. Figure 1e displays the intensity profiles of Mg, Al, and O elements with the corresponding STEM image. From the ratio of different elements, the chemical composition is confirmed to be consistent in the three MgAlO<sub>x</sub> barriers.

**Demonstration of the Formation of Double QWs in One MTJ.** Our first goal is to demonstrate the successful realization of DQW states in one MTJ structure. To do that, one has to be able to grow the QW structure on relatively thick tunneling barrier in order to measure a sizable tunneling resistance. In our previous work,<sup>25</sup> the QW layer has been grown on the thin 3 ML thick MgAlO<sub>x</sub> or MgO bottom barriers. This thickness was too thin to obtain a reasonable junction resistance even with a nanopillar structure. The bottom barrier thickness should be further increased. Nevertheless, in that case the QW states are found to be attenuated rapidly when increasing the thickness of the bottom MgAlO<sub>x</sub> layer (see Supporting Information S3). This could be due to the increase of the surface roughness with the thickness of the oxide layer, resulting in the decoherence of electron phase at the MgAlO<sub>x</sub>/Fe interface.<sup>22</sup> Fortunately, we have found that the 6 ML thick MgAlO<sub>x</sub> could be a good compromise to obtain both the QW states and a measurable resistance for the nanopillar junctions. On the basis of the aforementioned reasons, we have designed the triple-barrier MTJ (Sample I) to demonstrate the formation of the double QW states. The core structure comprises two thin barriers at both sides and one thick barrier in the middle separated by two Fe QW layers with identical thicknesses, as schematically shown in Figure 2a. In this structure, the resistance of the thin barrier is 100 times lower than that of the thick barrier so that most of bias voltage drop occurs within the thick barrier. The QW energy levels of the bottom and top Fe QW layers are almost coupled with the

Fermi levels ( $E_F$ ) of the under and upper bulk Fe electrodes, respectively. As shown in Figure 2b, when the negative bias is applied,  $E_F$  of the upper Fe electrode and the top QW energy levels are lifted up together. When the energy of electrons matches with the discrete QW states in the bottom QW layer under a specific bias, it results in the resonant tunneling and a local maximum of conductance. Thus, the oscillations of the conductance under the negative bias reflects the QW states formed in the bottom Fe QW layer. In a similar way, the QW states within the top QW layer can be detected by applying a positive bias, as schematically shown in Figure 2c. If the QW states can be formed in both QW layers, we expect to observe symmetric conductivity oscillation with bipolar bias (both positive and negative applied bias).

Figure 2d–g shows the transport properties of Sample I. The magneto-response TMR– $H$  curves at room temperature (RT) and 15 K are shown in Figure 2f. The magnetic field ( $H$ ) was applied along the Fe[100] easy axis direction and the TMR ratio is calculated by  $(R_{AP} - R_P)/R_P \times 100\%$ , where  $R_{AP}$  and  $R_P$  are the resistances of the antiparallel (AP) and parallel (P) magnetization configurations of the two adjacent QW layers, respectively. Because the middle thick barrier mainly dominates the junction resistance, the TMR ratio is indeed determined by the P and AP states of the two QW layers. From the TMR curve, it is found that the full AP state is difficult to reach due to the similar coercivity of the two Fe QW layers. Nevertheless, we still get a TMR ratio of 51% at RT due to the dipolar coupling between the two Fe QW layers within the nanopillar structure.<sup>32</sup> Remarkably, quite symmetric oscillations in the differential conductance are observed under negative and positive bias in both P and AP states, as shown in Figure 2d,e. The amplitude of oscillation is much stronger in P state than that in AP state, while the oscillation periodicity can be well matched in both magnetization configurations. The





**Figure 3.** (a) Schematics of the double QW MTJ with core structure of Fe/MgAlO<sub>x</sub>(6 ML)/Fe/MgAlO<sub>x</sub>(6 ML)/Fe/MgAlO<sub>x</sub>(6 ML)/Fe and the setup of measurement. (b,c) Schematics of resonant tunneling in the sample with bias  $V < 0$  and  $V > 0$ , respectively. The energy of electrons is partially thermalized before tunneling into the second QW. (d,e) Bias dependence of differential conductance for the junction in magnetic parallel and antiparallel states, respectively. Inset in (d) shows the energy positions of QW states and the fitting by PAM method. (f) TMR– $H$  curves of the junction at  $T = 15$  K and 300 K. (g) Bias dependence of differential TMR ratio at different temperatures.

oscillation is still detectable up to RT. The observation of conductivity oscillation in the AP state can be understood due to the band folding effect in the spinel MgAl<sub>2</sub>O<sub>4</sub> MTJ, resulting in the existence of  $\Delta_1$  conductive channel in the minority spin states, as demonstrated by the ab initio calculation.<sup>33</sup> In Figure 2g, the differential TMR (DTMR), calculated from the differential conductance defined as  $\text{DTMR} = [(dI/dV)_P - (dI/dV)_{AP}] / (dI/dV)_{AP} \times 100\%$ , shows clear oscillations with the same periodicity as those of the conductivity oscillations.

To prove that the oscillations originate from the formation of the QW states, we performed simulations using a phase accumulation model (PAM).<sup>34</sup> The PAM describes the quantization condition for the existence of a QW state as

$$2k_{\perp}d - \Phi_1 - \Phi_2 - \Phi_{\text{inf}} = 2\pi n \quad (1)$$

where  $k_{\perp} = \sqrt{2m^*(E - E_L)}/\hbar$  is the crystal momentum wave vector in the direction perpendicular to the interface,  $d$  indicates the Fe QW thickness, and  $\Phi_1 = \Phi_2 = 2 \sin^{-1} \sqrt{(E - E_L)/(E_U - E_L)} - \pi$  are the reflection phase shift at the two Fe/MgAlO<sub>x</sub> interfaces which confine the QW structure.  $m^*$  is the effective mass of majority  $\Delta_1$  electron in Fe, and  $E_L$  and  $E_U$  are the energies of the lower and upper edges of the barrier band gap. Here, we set  $m^* = 1$ ,  $E_L = -1.0$  eV and  $E_U = 3.9$  eV similar to ref 25. The parameter  $\Phi_{\text{inf}}$  taking into account the additional phase shift at interfaces, is set to 0 for simplification. By injecting all parameters into eq 1, we obtain the following expression

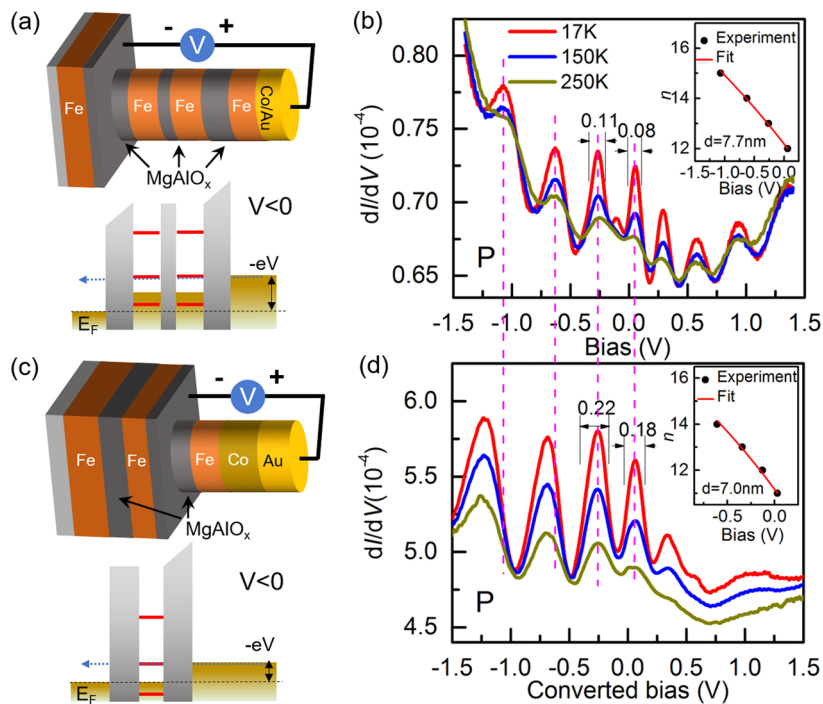
$$n = 1.63d\sqrt{V+1} - 0.637 \sin^{-1} \sqrt{\frac{V+1}{4.9}} + 1 - n_0 \quad (2)$$

where  $n_0$  is the index of the first QW state close to  $E_F$ . As displayed in the inset of Figure 2d, with  $n_0 = 15$  the positions

of the local conductance maximum under bipolar bias conditions can be well fitted with the PAM model with an identical QW thickness  $d = 9.0$  nm for both QWs. This clearly confirms the formation of double QW states in the triple-barrier MTJ Sample I.

**Resonant Tunneling in MTJ with QWs in Series.** To study the resonant tunneling through multi-QW structures, we have to float the energy levels in the QW and decouple the energy alignment between the QW states and  $E_F$  of thick upper and bottom Fe electrodes. For this purpose, we have designed Sample II with three MgAlO<sub>x</sub> barriers with identical thickness of 6 MLs while the thickness of top and bottom QWs is kept different (9 and 7.7 nm, respectively), as schematically shown in Figure 3a. In this sample, the voltage is dropped almost equally on the three barriers. Figure 3d–g shows the transport properties of Sample II. The TMR– $H$  curves at RT and 15 K are shown in Figure 3f. The clear resistance plateaus in the low magnetic field range indicates the establishment of a full AP state in this sample, which could be due to the enhanced AP dipolar coupling not only between the two Fe QW layers but also between the Fe QW layers and the thick Fe electrodes after the nanopillar structure formation. As a consequence, we have measured a higher TMR ratio of 110% at 15 K and 85% at RT.

Figure 3d,e shows the bias dependence of the differential conductance in P and AP states, respectively. Clear oscillations are found for both positive and negative biases, but the periodicity is different. Normally, one would expect that the tunneling of electrons could have been influenced by the two QWs' states in series and thus two sets of conductance oscillation under one polar bias could be observed when the injector's  $E_F$  is aligned with the QW states in the two wells. However, our results clearly show that only one set of oscillation corresponding to the states in one QW is present.



**Figure 4.** (a) Schematics of double QW MTJ with core structure of Fe/MgAlO<sub>x</sub>(6 ML)/Fe/MgAlO<sub>x</sub>(3 ML)/Fe/MgAlO<sub>x</sub>(6 ML)/Fe and schematics of the coherent resonant tunneling through double QWs with bias  $V < 0$ . (b) Bias dependence of differential conductance for the DQW junction represented in panel (a) in magnetic parallel state. (c) Schematics of single QW MTJ with core structure of Fe/MgAlO<sub>x</sub>(6 ML)/Fe/MgAlO<sub>x</sub>(6 ML)/Fe and schematics of the resonant tunneling in the junction with bias  $V < 0$ . (d) Bias dependence of differential conductance for the SQW junction shown in panel (c) in magnetic parallel state. The bias is multiplied by a factor of 2 in order to compare with panel (b). Please see details in the main text. Insets of (b,d) show the energy positions of QW states and the fitting by PAM method.

From the periodicity of the oscillation, it is reasonable to suppose that the conductance oscillations at positive and negative bias are attributed to the top and bottom QW states, respectively, due to their different QW thickness. The possible explanation is that the electrons are rapidly dephased and thermalized<sup>35,36</sup> inside the first QW before they can resonantly tunnel into the second QW, as schematically drawn in Figure 3b,c. In semiconductor QW, hot electrons typically thermalize in a picosecond time scale, dissipating energy via various phonon-mediated relaxation pathways.<sup>36</sup> In metallic QW, electron thermalization is even enhanced due to the smaller electron mean-free path compared to the semiconductor. Therefore, the electrons will quickly lose their phase and energy coherence if they are confined in a QW for a longer time and the reception side with another QW structure has no available resonant QW energy level under that bias. In the end, the MTJ with QWs in series only reveals the resonant tunneling characters of the last QW which the electrons tunnel through.

To further confirm our supposition, we have performed the analysis of the periodicity with the PAM model. To correctly take into account the electron thermalization effect, we have added a factor of  $F$  in eq 2

$$n = 1.63d\sqrt{V^*F + 1} - 0.637 \sin^{-1} \sqrt{\frac{V^*F + 1}{4.9}} + 1 - n_0 \quad (3)$$

This  $F$  factor is introduced to represent the true energy of electrons after tunneling through the first QW to probe the unoccupied QW states in the second QW. As shown in the inset of Figure 3d, with the factor  $F = 0.56$ , the periodicity at positive and negative bias can be well fitted with the states in

the 7.3 and 8.7 nm thick QWs, respectively, which well correspond to the top and bottom QWs. In the situation of a full thermalization down to  $E_F$  in the first QW, one should obtain a  $F = 1/3$  by taking into account the voltage drop on one barrier. On the contrary, if the electrons maintain their energy when tunneling into the second QW,  $F$  factor should be equal to  $2/3$  due to the voltage drop on the two barriers. Therefore, our fitted factor  $F = 0.56$  between  $1/3$  and  $2/3$  indicates that the electrons are partially thermalized in the first QW before tunneling into the second QW. One remarkable feature in this kind of DQW MTJ is that with the design of different thickness for the two QWs, one can achieve different oscillation periodicities with bipolar bias, which could be of interest to generate new functionality with the multi-QW MTJs.

Figure 3g displays the DTMR curves calculated from the different conductance in P and AP states. Although clear oscillations can be observed in both positive and negative bias, the amplitude of oscillation is smaller than that of Sample I. This can be understood by noticing that the influence of junction resistance in Sample II due to the QW resonant tunneling only occurs in one QW, while the total junction resistance should take into account three tunneling barriers. Therefore, the oscillation amplitude is reduced by a factor of 3 compared to Sample I.

**Evidence of Coherent Resonant Tunneling through Double QWs.** As mentioned above, to achieve the coherent resonant tunneling through two QWs, one must accelerate the tunneling of electrons through QWs and minimize their energy thermalization inside the QWs. To fulfill the conditions for such kind of coherent resonant tunneling, the discrete QW energy levels in the two QWs should be aligned under a certain

bias. The following three aspects were considered to design such DQW structure. First, the two QW layers must have an identical thickness so that their whole set of QW states can be matched with each other at zero bias to fulfill the resonant tunneling condition. Second, the QW states should be decoupled with the  $E_F$  of thick electrodes so that one can adjust the bias to probe different unoccupied QW states. Third, to ensure the energy alignment between the two QWs' states, the voltage drop between the two QWs should be minimized. On the basis of these considerations, we have designed a DQW MTJ structure (Sample III) with two thick barriers (6 MLs) at both sides and one thin barrier (3 MLs) in the middle, as schematically shown in Figure 4a. The two QW layers have identical thickness of 7.7 nm. The difference in resistance between the thick and thin barriers ( $\sim 100$  times) ensures that the main voltage is dropped on the two thick barriers and very small bias is applied across the two QWs when performing bias dependence measurements. Figure 4a schematically shows the diagrams of the resonant tunneling through the two QWs under a negative bias.

Figure 4b shows the bias dependence of the differential conductance in P state. Oscillations in conductance can be clearly observed in the bipolar bias condition. However, the amplitude of oscillation under negative bias is stronger than that under positive bias. This could reflect a certain asymmetric growth condition for the MgAlO<sub>x</sub>/Fe interface and the Fe/MgAlO<sub>x</sub> interface.<sup>14</sup> To prove that the observed quantum oscillation originated from the coherent resonant tunneling between the two QWs, we have fabricated one reference sample (Sample IV) with SQW structure [MgAlO<sub>x</sub>(6 MLs)/Fe(7.7 nm)/MgAlO<sub>x</sub>(6 MLs)], which is schematically shown in Figure 4c. In this sample, because the etching is stopped in the top thick MgAlO<sub>x</sub> barrier the applied bias is totally dropped across the top barrier due to its dominated resistance in the junction. In comparison, in the DQW sample only half of the bias is applied across one thick barrier to probe the unoccupied QW states. Therefore, to quantitatively compare the QW state positions for both samples, we have intentionally multiplied by a factor of 2 for the bias voltage of Sample IV. Figure 4d shows the parallel conductance as a function of the converted bias voltage for Sample IV. Strong oscillations can be observed only for negative bias voltage. This is because that the Fermi level  $E_F$  of the bottom electrode cannot be lifted to probe the unoccupied QW states under the positive bias due to the small resistance of the bottom barrier with large micron size of area. Remarkably, the four main peaks (indicated with dash lines) at negative bias correspond perfectly to those of Sample III. The slight difference in position is enlarged for the QW states at larger bias, which is due to a smaller QW thickness in Sample IV [7.0 nm estimated by PAM simulation in the inset of Figure 4d]. This gives a strong argument that although the DQW structure possesses double 7.7 nm thick QWs, it really acts as the same behavior of a single 7.7 nm thick QW. The QW state positions can be well fitted in the inset of Figure 4b. This also proves that the electrons can coherently resonantly tunnel through the two QWs without energy thermalization in one of the QWs. These results would not have been observed if the electrons had thermalized in the QWs. The situation would have been similar to what we have observed for the SQW reference sample under positive bias condition.

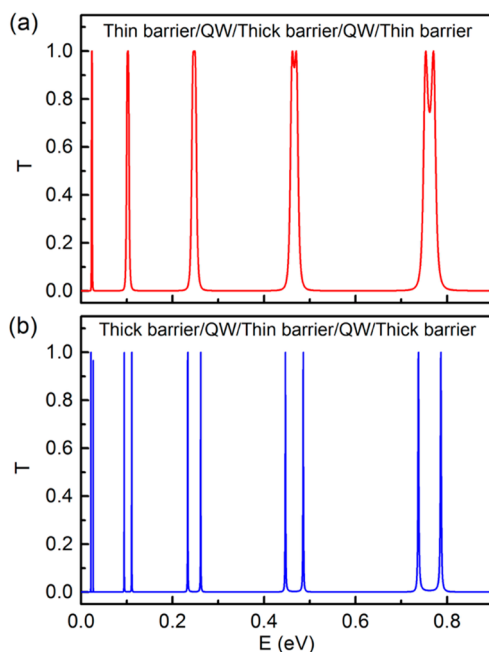
To enhance coherent resonant tunneling through the two QWs, two conditions are crucial. Indeed both the thin middle

barrier and the available energy levels in the second QW must ensure the electrons rapid tunneling into the second QW states without too much time dwelling in the first QW. Another remarkable feature for such coherent resonant tunneling appears in the resonant peak width. As clearly seen in Figure 4b,d, the width of the oscillation peak for the DQW Sample III is about half of SQW Sample IV. Although the QW states act as an energy filter, double QW structure can further enhance this energy selection effect for the tunneling electrons, which results in narrower oscillation peak width. This finding also reinforces our conclusion that the observed oscillation features in Sample III originated from the coherent resonant tunneling through the double QWs. One could also expect that the DQW must have stronger oscillation amplitude than that of SQW. Indeed the QW states in Fe are formed by majority  $\Delta_1$  electron band.<sup>4,8</sup> The QW can select not only the energy of tunneling electrons but also the Bloch state symmetry feature linked to the band structures. Therefore, the DQW can effectively suppress the conduction channels from other electron bands than the majority  $\Delta_1$  band, resulting in an enhanced oscillation amplitude. However, the oscillation amplitude could also be influenced by many other factors, such as QW thickness and the etching stop layer (see Supporting Information S2). A slight change in the QW layer thickness and the different etching stop layer in the SQW sample could effectively increase the oscillation amplitude and make the direct comparison with the DQW sample difficult. Undoubtedly, future reduction of the QW thickness in the DQW structure should exponentially increase the oscillation amplitude.

It is well-known from quantum theory that in the multi-QW system, when the middle barrier is sufficiently thin, the two sets of QW states in the two wells can hybridize with each other and create a new set of QW states.<sup>37</sup> For a qualitative description of such physics in our DQW structure, we calculated the tunneling ratio of MTJ by using classic transfer matrix method on two DQW structures: thin barrier/QW/thick barrier/QW/thin barrier and thick barrier/QW/thin barrier/QW/thick barrier. Please see details of the calculation in Supporting Information S4. Figure 5 displays the calculated energy dependence of the tunneling ratio ( $T$ ) in the two kinds of DQW junctions. Evidently, the DQW junctions with three tunnel barriers exhibit peak splitting effect due to the hybridization between the QW states in the two wells. In the thin barrier/QW/thick barrier/QW/thin barrier junction (Figure 5a), the QW states weakly interact with each other and the splitting appears clearly at high bias voltage when the electron waves start overlapping through the barriers. On the contrary, in the thick barrier/QW/thin barrier/QW/thick barrier junction (Figure 5b), QW states strongly interact with each other and one QW state can be completely split into two states. Generally, the splitting of the satellite peaks can be controlled by the thickness of the central barrier. In our experiments (Figure 4b), the splitting in the resonant peaks cannot be detected. This could be due to two reasons. Either the middle barrier could be still too thick so that the hybridization of QW states is too weak. Or the energy distribution of tunneling electrons is too large and the splitting of peak is submerged by the overlap of two satellite QW states. In the future, the optimization of middle barrier thickness will allow us to observe such hybridization between two QWs.

In summary, the spin-dependent resonant tunneling properties have been studied in full epitaxial MgAlO<sub>x</sub> MTJ with





**Figure 5.** Simulation of energy dependence of tunneling ratio for double QW MTJs with structure of (a) thick barrier/QW/thin barrier/QW/thick barrier and (b) thin barrier/QW/thick barrier/QW/thin barrier, respectively.

double quantum well structures. Strong oscillations in conductance have been observed in the nanopillar structured DQW MTJ. It is found that the middle barrier plays a critical role on the resonant tunneling through the two QWs in series. When it is too thick, the electrons are partially thermalized in the first QW, and consequently the periodicity of the conductance oscillations are only determined by the second QW. When the resonant tunneling condition is fulfilled, the DQW MTJ can produce strong conductance oscillations with much narrower peak width (about half) owing to the enhanced energy selection effect compared to the SQW MTJ. The realization of multiple QWs in single MTJ paves an efficient way to engineer quantum resonant tunneling to control the magnetoresistance with bias in addition to the conventional magnetization manipulation.

**Methods. Sample Preparation.** The fully epitaxial MTJs with different configurations of  $\text{MgAlO}_x$  barriers were grown by MBE on  $\text{MgO}(001)$  substrate. First, the substrate was annealed at  $650^\circ\text{C}$  for 60 min and 10 nm  $\text{MgO}$  seed layer was deposited at RT to prevent the carbon diffusion during subsequent annealing process.<sup>38</sup> Then 45 nm Fe was grown at RT as the bottom electrode and annealed at  $500^\circ\text{C}$  for 30 min to improve the crystalline quality and flatten the surface. Double QW structures of  $\text{MgAlO}_x/\text{Fe}/\text{MgAlO}_x/\text{Fe}/\text{MgAlO}_x$  with different thicknesses of barriers and wells were then deposited, where each Fe layer was annealed at  $400^\circ\text{C}$  for 20 min. The layer-by-layer growth of the  $\text{MgAlO}_x$  barriers at RT was monitored by the oscillation of in situ RHEED intensity. Finally, the samples were capped with  $\text{Fe}(10\text{ nm})/\text{Co}(20\text{ nm})/\text{Au}(10\text{ nm})$  as the top electrode, where only the Fe layer was annealed at  $400^\circ\text{C}$  for 20 min. The Co layer is used to enhance the coercivity of the top Fe and the Au layer is used to prevent the sample from oxidation. The multilayers were patterned into nanopillar junctions with elliptical size of  $120 \times 240\text{ nm}^2$  by E-beam lithography combined with Ar ion etching

technique. The etching process was monitored by secondary ion mass spectrometry and can be precisely stopped in the desired  $\text{MgAlO}_x$  barrier.

**TEM Characterization.** Thin lamella TEM samples were prepared by focused ion beam (FIB) lift up method (FEI-FIB-Helios). Ex situ microstructural characterization was performed with probe aberration-corrected STEM microscopes (FEI-Titan-Themis) operated at 300 kV equipped with the super X energy dispersive X-ray spectrometer.

## ■ ASSOCIATED CONTENT

### 📄 Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.nanolett.9b00205.

In situ RHEED characterization; effect of etching stop in different barriers; effect of bottom barrier thickness on quantum well states; hybridization-induced splitting of energy level between two QWs; additional figures (PDF)

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### Notes

The authors declare no competing financial interest.

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